# Structural and magnetic properties of Nanocrystalline Nickel ferriteprepared by Citrate Sol—geland Solid- state reaction techniques

# R.M.Hamdy<sup>1</sup>, M.B.S.Osman<sup>1</sup>, Y.M.abbas<sup>2</sup>, M.A.Ahmed<sup>3</sup>

#### Abstract

The structure and magnetic properties ofnanocrystalline nickel ferrite powder NiFe<sub>2</sub>O<sub>4</sub> has been investigated using two different preparation methods, including the ceramic technique and citrate method. The synthesized powders were characterized using X-ray Diffraction (XRD) for crystallite size, X-ray density and lattice parameter calculation. The results indicated that the citrate method gives the lowest value for the lattice parameter and crystallite size (60.6 nm) in citrate method and (73.8 nm) in ceramic method. Distribution of cations among the two interstitial sites (tetrahedral and octahedral sites) has been estimated by analyzing the powder X-ray diffraction patterns by employing Rietveld refinement technique, and the results reveal the existence of samples asinverse spinelwith cubic structure and Fd-3m space group. The morphological investigations using Field Emission Scanning Electron micrograph (FE-SEM) and High Resolution Transmission Electron Microscopy (HR-TEM). The elemental analysis of samples using Energy Dispersive X-ray Analysis (EDAX) . Magnetic measurements of the samples at room temperature were carried out by means of vibrating sample magnetometer (VSM).

**Key words**: Nickel ferrite, Rietveld, Cation distribution

#### 1. Introduction

The ferrite of greatest technical importance is derived crystallographically from three natural compounds: the spinel, the garnet, and the magentoplumbite. Spinel is an important class of mixed-metal oxides, which has the structure of the natural spinel MgAl<sub>2</sub>O<sub>4</sub> and the general chemical composition of AB<sub>2</sub>O<sub>4</sub>, A and B can be divalent, trivalent cations. The physical properties of the spinel ferrites are controlled by the choice of the cations and their distribution between tetrahedral and octahedral sites of the structure. Nickel ferrite is one of the multifaceted and technologically important soft ferrite materials because of its typical ferrimagnetic properties, lower eddy current losses, low conductivity and high electrochemical stability[1-2]. Nickel ferrite is veryattractive in recent research works because of its higherelectromagnetic performance,

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excellent chemical stability, mechanical hardness, high coercivity, and moderate saturation magnetization [3].

Nickel ferrite (NiFe<sub>2</sub>O<sub>4</sub>) belongsto the inverse spinel. In this structure the 32 divalent oxygen ions formed a closed packed FCC arrangement with 64 tetrahedral interstitial A-sites and 32 octahedral B-sites. Out of these, the divalent (Ni<sup>2+</sup>)and trivalent (Fe<sup>3+</sup>) cationsoccupy one-eighth of A-sites and half of the B-sites [4]. Among these nanomaterials, the magnetic nanomaterials are more interesting owing to their medical, electronic and recording applications. These applications depend on the size, shape, purity and magnetic stability of these materials [5] such as wave absorber, magnetic core memory devices and carriers of drug inside the body [6] this paper compares between the synthesis of NiFe<sub>2</sub>O<sub>4</sub> nanoparticles which prepared by ceramic technique and citrate route, The results for the structural and elemental analysis by XRD, EDAX, FE-SEM and HR-TEM and magnetic properties characterized by VSM are discussed.

#### 2. Experimental

## 2.1 Synthesis techniques

## (1) Ceramic precursor: sample S1

The starting materials were stoichiometric quantities of analytical reagent grade powders of Fe<sub>2</sub>O<sub>3</sub> and NiO; all 99.9 % pure, supplied by LobaChemie used for the polycrystalline samples of Nickel ferrite. They were first the stoichiometric mixtures of all the ingredients were thoroughly mixed in wet medium (acetone) and then grinded for 5hours in an agate mortar to obtain a very fine powder of good homogeneity. The mixture was then presented at 700°C for 6 h at a heating / cooling rate 5 degrees/min in high purity crucibles .The presintering process to ensure that all the carbon is liberated from the mixture in the form of CO<sub>2</sub>. The presintered mixture was again well grinded for 15min then pressing by the hydraulic press to pellets and resulting were finally sintered at 1100°C for 6h.we denote this sample as S1.

#### (2) Citrate precursor: sample S2

The starting materials were analytical reagent grade powders of Fe (NO<sub>3</sub>)<sub>3</sub>.9H<sub>2</sub>O and Ni(NO<sub>3</sub>)<sub>2</sub>.6 H<sub>2</sub>O; all 99.9 % pure. The molar ratio of metal nitrates to citric acid was taken as 1:1 molar ratio. The metal salts and citric acid dissolved in minimum amount of de-ionized water and PH was adjusted to 7 by using ammonia. Then the mixture was heated on the hot plate, the decomposition reaction would not stop before the whole citrate complex was consumed. The auto-ignition was completed within a minute, yielding the brown- colored ashes termed as a precursor. The as-prepared powder of the sample was annealed at 900 °C for 4 h with rate 5degree/min.we denote this sample as S2.The flow chart of the synthesis route of eachmethod is shown in fig. (1).

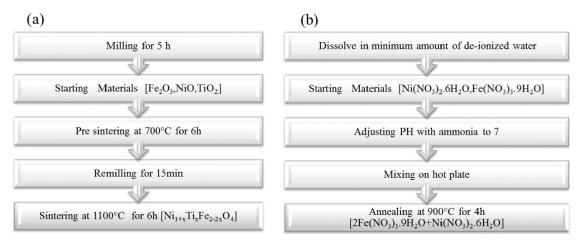


Fig.(1). flowchart for the synthesis of NiFe<sub>2</sub>O<sub>4</sub> by (a) ceramic method (b) citrate method.

#### 2.2Characterizationoftheassynthesizedmaterials

The crystalline phase of all the samples was identified at room temperature using PhilipsX-ray Diffraction Instrument with X'pert MPD Diffractometer,with  $CuK_{\alpha}$  radiation ( $\lambda$ = 1.54060 °A). All the X-ray diffraction (XRD) patterns were analyzed with the help of FullProfprogramme by employing Rietveld refinement technique. The XRD patternsfor all the samples could be refined using the Fd-3m space group [7]. The Rietveld method is awell-established technique for extracting structural details from powder diffraction data. The method employs a least-squares procedure to compare Bragg intensities and those calculated from a possible structural model. In the first step of refinement, the global parameters, such as background and scale factors, were refined. In the next step, the structural parameters such as lattice parameters, profile shape and width parameters, preferred orientation, asymmetry, isothermal parameters, atomic coordinates, and site occupancieswere refined in sequence. The fitting quality of the experimental data is assessed by computing the parameters such as the 'goodness of fit'  $\chi$ 2 and the R factors (R p=profile factor, R<sub>B</sub> = Bragg factor, and R<sub>f=</sub>crystallographic factor) [8]. When these parameters reached their minimum value, the best fit to the experimental diffraction data is achieved. We have calculated the crystallite size by The Scherrer's formula [9] is defined as

D= 
$$k\lambda/\beta$$
 Cos θ

where Scherrerconstant k depends upon the shape of the crystal (which is equivalent to 0.9, assuming the spherical grain),  $\beta$  is the full width at half maximum of the intensity (in radian) vs. 20 profile,  $\lambda$ is the wavelength of the  $CuK_{\alpha}$  radiation (equal to 0.1540 nm),  $\theta$  is the Bragg's diffraction angle, and D is the crystallite size. In Scherrer's formula, the average crystallite size has been calculated using Gaussian fit, fitted to the peaks in XRD pattern. D has been taken average for all the peaks.

The X-ray density was calculated using the molecular weight and the lattice constant. The XRDdensity was calculated by formula [10]

$$\rho_{XRD} = 8M/Na^3$$

Where M is molecular weight of the sample and N is the Avogadro number (equal to  $6.022140857 \times 10^{23}$ ) and 'a' is the lattice parameter.

The morphology, structure and elemental composition of NiFe<sub>2</sub>O<sub>4</sub> were characterized by Field-emission Scanning electron microscopy (FE-SEM) (QUANTA FEG 250) and Energy dispersive X-ray analysis (EDAX) were performed to determine the Chemical composition of the elements present from the surface to the interior of the solids and to confirm the homogeneity of the as prepared samples. High-resolution transmission electron microscopy (HR-TEM) analysis was performed in a JEOL (JEM-2100); HR-TEM micrographs were used to characterize the particle size, shape, grain size and to confirm the lattice parameter variation across the interfaces using selected area electron diffraction (SAED) pattern.

## 2.3 Magnetic study

The magnetic properties of the investigated solids were measured at room temperature using a vibrating sample magnetometer (VSM; Lake Shore, Model-7410). From the obtained hysteresis loops, the saturation magnetization ( $M_s$ ), remanence magnetization ( $M_r$ ), coercivity ( $H_c$ ) and Squareness ratio were determined.

#### 3. Results and discussion

# 3.1 XRD analysis

The X-ray diffraction pattern study for the nickel ferrite nanoparticles synthesized by ceramic and citrate precursor was done to identify the possible formation of single phase. The diffraction peaks for the synthesized nickel ferrite nanoparticles samples correspond to well indexed reflection planes (111), (2 2 0), (3 1 1), (2 2 2), (4 0 0), (4 2 2), (5 1 1), (4 4 0) and (533) are shown in Fig.1, which thisplanes indicating the cubic spinel structure of nickel ferrite. No impurity peaks are observed in the XRD spectra of as synthesized samples. The average particle size of the nickel ferrite nanoparticles samples calculated from the Scherrer's formula, the lattice parameter and The X-ray density is shown in Table 1.

Table 1 XRD parameters of NiFe<sub>2</sub>O<sub>4</sub>

	Sample S1 NiFe <sub>2</sub> O <sub>4</sub> :ceramic method	Sample S2 NiFe <sub>2</sub> O <sub>4</sub> :citrate Method		
D(nm)	73.8	60.6		
$\rho_{\rm XRD}  ({\rm g/cm}^3)$	5.20	5.37		
a (°A)	8.337	8.335		

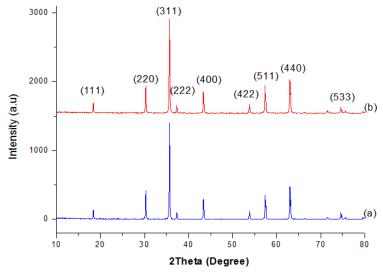


Fig.1 powder XRD patterns of NiFe<sub>2</sub>O<sub>4</sub> nanoparticles prepared by (a) Ceramic route (sample S1) (b) Citrate route (sample S2)

The X-ray diffraction patterns along with Rietveld refined data have been shown in Fig.2a,b for the samples S1 (prepared by the ceramic precursor method) and S2 (prepared by the citrate precursor method), respectively. The allowed Braggs positions for the Fd-3m space group are marked as verticallines. We have observed that all the experimentalpeaks are allowed Bragg 20 positions for Fd-3m space group. In the refinement, the oxygen positions (x = y = z) have been taken as free parameters. However, all other atomic fractional positions have been taken as fixed. Other parameters such as lattice constants, isothermalParameters, occupancies, scale factors, and shape parameters have been taken as free parameters. The background has been corrected by pseudo-Voigt function. Typical fractional positions of the atoms for samples S1 and S2 are given in Table 2. The refined XRD patterns show that the samples are in single phase form. The various R factors are listed in Table 2. We have observed a lowvalue of  $\chi^2$  (goodness of fit) which justifies the goodness of refinement. The refined lattice parameters and unit cell volumes for Samples S1 and S2 have been listed in Table 3. Bond lengths and bond angles were calculated using refined lattice parameters and fractional coordinates with the help of Diamond programme. Typical unit cell showing bond length between cations and anions for Samples S1 and S2 has been depicted in fig 3a, b. The values of bond lengths and bond angles for samples S1 and S2 are listed in Table 4.

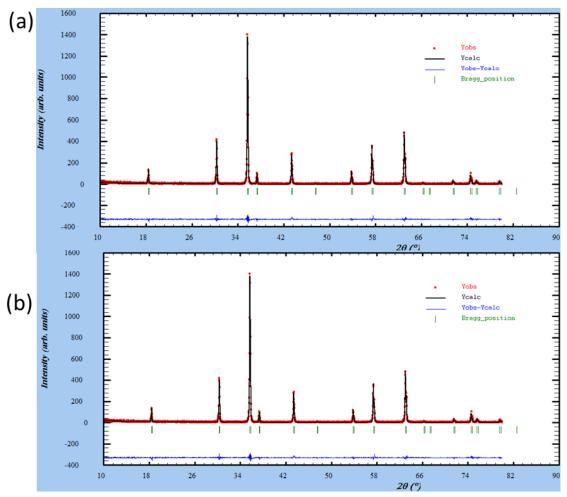


Fig.2 Rietveld refined XRD patterns for samples (a) S1 (NiFe<sub>2</sub>O<sub>4</sub>:ceramic method) (b) S2 (NiFe<sub>2</sub>O<sub>4</sub>:citrate method)

We have observed that sample S1 has a larger crystallite size than sample S2. Although both samples were prepared under identical conditions, the crystallite size was not the same. The crystallite size is related to the relative

Inter dependence between the nucleation and growth steps, and it is strongly affected by the synthesis route. Thus, the citrate precursor method makes the formation of crystallites easier due to suitable growth of the crystals.

One can see that Ni ions are present on both tetrahedral and octahedral sites and Fe ions are present on both tetrahedral and octahedral, which reveals that the samples are in inverse spinel structure irrespective of the preparation method.

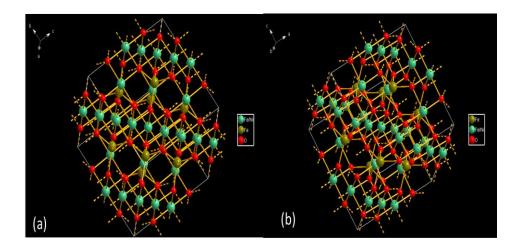
Table 2Typical atomic coordinate (x, y, z) of different atoms for samples S1 and S2

Method	Atoms	Х	у	Z
eramic method				
	Fe	0.12500	0.12500	0.12500
	Ni/Fe	0.50000	0.50000	0.50000
	O			0.25450
		0.25450	0.25450	
	Fe Ni/Fe O	0.12500 0.50000	0.12500 0.50000	0.12500 0.50000 0.25522
trate method		0.25522	0.25522	

Table 3Rietveld agreement factors, lattice constant, and unit cell volume of samples S1 and S2 at  $\,$ 

	Sample S1 NiFe <sub>2</sub> O <sub>4</sub> :ceramic	Sample S2 NiFe <sub>2</sub> O <sub>4</sub> :citrate
	method	Method
R <sub>p</sub> (%)	18.3	19.6
R <sub>B</sub> (%)	3.37	3.97
R <sub>f</sub> (%)	2.50	2.77
R <sub>f</sub> (%) χ <sup>2</sup>	0.844	1.07
a=b=c (° <b>A</b> )	8.3371	8.3359
V (°A) <sup>3</sup>	579.498	579.249

 $R_p$ , profile factor;  $R_B$ , Bragg factor;  $R_f$ , crystallographic factor;  $\chi^2$ , goodness of fit factor; a, lattice constant; V, unit cell volume.



 $Fig. 3 \ The \ crystal \ unit \ cell \ of \ NiFe_2O_4(a) \ sample \ S1(ceramic \ method) \ (b) \ sample S2(citrate \ method)$ 

# Table 4Bond lengths and Bond angles between atoms of Sample S1 NiFe $_2\mathrm{O}_4$ and Sample S2 NiFe $_2\mathrm{O}_4$

Sample S1 NiFe <sub>2</sub> O <sub>4</sub>							
Atom 1	Atom 2	d 1,2 [°A]	Atom 3	d 1,3	Angle		
				[°A]	2,1,3		
Fe Ni	0	2.0475	0	2.0475	87.881		
	0	2.0475	0	2.0475	92.119		
	0	2.0475	0	2.0475	179.988		
Fe	0	1.87	0	1.87	109.471		
0	Fe	1.87	Fe Ni	2.0475	123.779		
	Fe Ni	2.0475	Fe Ni	2.0475	92.08		
Sample S2 NiFe <sub>2</sub> O <sub>4</sub>							
Atom 1	Atom 2	d 1,2 [°A]	Atom 3	d 1,3	Angle		
				[°A]	2,1,3		
Fe Ni	0	2.0414	0	2.0414	92.468		
	0	2.0414	0	2.0414	87.532		
	0	2.0414	0	2.0414	180		
Fe	0	1.8802	0	1.8802	109.471		
0	Fe	1.8802	Fe Ni	2.0414	123.537		
	Fe Ni	2.0414	Fe Ni	2.0414	92.416		

## 3.2 Field-Emission Scanning Electron micrograph with EDAX analysis:

The surface morphology characteristics of the synthesized nickel ferrite nanoparticles synthesized by ceramic and citrate method were investigated using (FE-SEM). Fig. 4a, cclearly shows the cubic shape of nickel ferrite nanoparticles and the morphology of the particles isvery similar. It also indicates the polyhedral shape and narrow size distribution of the particles. As the sample synthesized by citrate results in a very fine powder, so there is formation of soft agglomeration between particles of Nickel ferrite.

Fig. 4b, d shows the EDAX pattern for the elemental analysis of nickel ferrite nanoparticles. The elemental analysis confirms the homogeneous mixing of Ni, Fe and O atoms. As there is no impurity peaks observed in the EDAX spectrum except the extra gold peak. The presence of gold peak is due to the thin coating on the sample surface to make it conducting, which is required to record the FE-SEM. The compositional stoichiometry of the Nickel ferrites observed by EDAX spectra are in good agreement with the stoichiometric calculation.

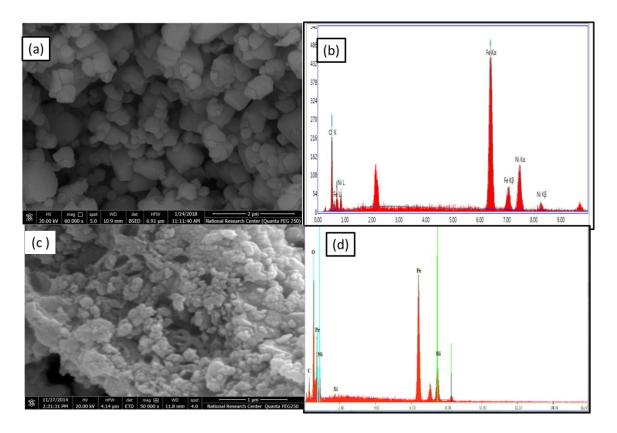


Fig.4 (a) FE-SEM (b) EDAX patterns of NiFe<sub>2</sub>O<sub>4</sub>(ceramic method), (c) FE-SEM (d) EDAX patterns of NiFe<sub>2</sub>O<sub>4</sub>(citrate method)

# 3.3 High Resolution Transmission Electron Microscope (HR-TEM) with SAED

HR-TEM helps to visualize the inherent matrix of individual particles size, morphology and structure of nickel ferrite nanoparticles. Fig.5a, c depicts the HR-TEM images of the NiFe<sub>2</sub>O<sub>4</sub>synthesized by ceramic and citrate method which confirms the clear morphology of NiFe<sub>2</sub>nanoparticles. The image reveals that the synthesized NiFe<sub>2</sub>nanoparticles are cubic in shape.

The selected area electron diffraction (SAED) pattern of the cubic shaped NiFe<sub>2</sub>O<sub>4</sub> nanoparticles is shown in Fig.5b, d.All spots are identified as the diffractions from cubic NiFe<sub>2</sub>O<sub>4</sub>, which reveals the crystalline nature of NiFe<sub>2</sub>O<sub>4</sub>nanostructure.

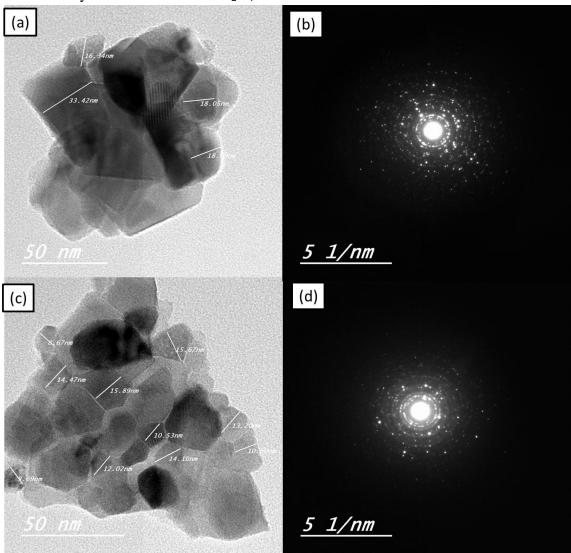


Fig.5 (a) HR-TEM (b) SAED patterns of sample S1, (c) HR-TEM (d) SAED patterns of sample S2

## 3.4. VSM analysis

Magnetic properties of the synthesized nickel ferrite nanoparticles sample is characterized by VSM (Vibrating sample magnetometer) at room temperature 300°K. Shape and size of the nanoparticle highly affectsThe magnetic properties of the material [11]. Fig. 6 shows hysteresis loop obtained is "S" shape curve that showsthe sample prepared by the citrate method has superparamagnetic behavior, Table 5depicts the values ofCoercivity (Hc), saturation magnetization(Ms) ,Squareness ratio (M<sub>r</sub>/M<sub>s</sub>) andRetentivity (M<sub>r</sub>). Which is significantly high as compared to the saturation magnetization of the bulk nickel ferrite (50.4 emu/g)[12]and reported value for the multi domain bulk nickel ferrite (55 emu/g). Increased in the value of Hc can be ascribed to the grain growth. The calculated values of squareness ratio of the sample are below 0.5 clearly shows that thenickel ferrite nano-particles synthesized through ceramic and citrate method are multi-domain in nature.

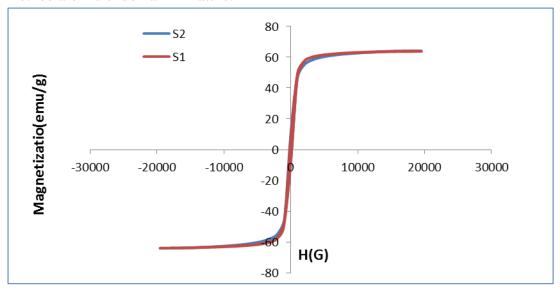


Fig.6 Magnetic hysteresis curve for nickel ferrite nanoparticles measured at room temperature.

Using the value of saturation magnetization at room temperature, the magnetic moment per formula unit in Bohr magneton  $\mu_B$  is calculated by the following equation [13]:

$$n_{\rm B} = \frac{\rm Mwt \times Ms}{5585} (1)$$

The magnetic moment per formula unit in Bohr magneton was also calculated according to the Neel's ferrimagnetic theory as,

$$n_B = M_B - M_A(2)$$

Where  $M_B$  and  $M_A$  are the magnetic moments in Bohr magneton ( $\mu_B$ ) for A and B sites. The magnetic moments for Fe<sup>3+</sup> and Ni<sup>3+</sup>ions are 5 and  $2\mu_B$ , respectively, which are used in the present calculations. Theresults are summarized in Table 5.

Table 5 Magnetic properties of synthesized sample-S1 (ceramic method) and sample-S2 (citrate method).

Sample	M <sub>r</sub> (emu/g)	M <sub>s</sub> (emu/g)	$M_r/M_s$	H <sub>c</sub> (G)	$n_{\mathrm{B}}$	
					Cal.(XRD)	Obs.(VSM)
S1	7.7215	64.122	0.12042	137.82	2	2.690
S2	10.819	64.148	0.16866	10.819	2	2.692

#### 4. Conclusion

 $1\text{-NiFe}_2O_4$  ferrites have been successfully synthesized by ceramic and citrate method and successful formation of cubic spinel structure with F d -3 m space group confirmed by X-ray diffraction analysis. The crystallite size, the lattice parameter and x-ray density were found using the XRD parameters. There is no difference between the preparation methods in

That they all reveal the single phase with cubic structure.

- 2- The value of lattice parameter is highest in the ceramicmethod while its value decreases in all the wet methodsdepending on the particle size and the melting point of theraw materials. The sample prepared by the citrategivesnanometric crystal size.
- 3-FE-SEM micrographs and EDAX analysis confirms the chemical compositions, which support our observations on the structure of the ferrite.
- 4- The Magnetic hysteresisloop of nickel ferrites confirms the formation of superparamagnetic behavior.

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# الملخص باللغة العربية

الخواص التركيبية والمغناطيسية لمتراكبات النيكل فيريت النانومترية المحضرة بطريقة تفاعل الحالة الخواص الصلبة و سيترات السائل الجيلاتيني

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# ملخص البحث:

تناولت الدراسة الحالية تحضير نظام من حديديات النيكلوله معادلة عام( $Ni_{1+x}Fe_{2-2x}O_4$ )باستخدام طريقتي تحضير الطريقة الاولي وهي طريقة السيراميك من اكاسيد الاملاح و حرقها عند درجات حرارة مختلفة، و الطريقة الثانية هي طريقة الاحتراث الذاتي لنترات المعادن بستخدام حمض الستريك.

وقد اجريت علي العينات التي تم تحضيرها القياسات المختلفة باستخدام وسائل معملية مثل حيود الاشعة السينية وقياسات المغناطيسية والميكروسكوب الالكتروني (الماسح و النافذ), كما تمت دراسة التركيب البلوري للعينات.

فأظهرت النتائج ان العيناتتتبع النظام المنتظم (النظام المكعبي متمركز الوجه) وتبين من تحليل النتائج باستخدام طريقتي لابي و ريتفلدكاتيونات النيكل تقوم باحلال كاتيونات الحديد في الموقع الرباعي (A) و تم رسم التركيب البلوي للعينات عن طريق برنامج .Diamond programme

واظهرت نتائج الميكرسكوب الالكتروني ان العينات تتكون من جسيمات في نطاق النانومتري وتتبع النظام المكعبي و تم دراسة الخصائص السطحية للعينات عن طريق برنامج Gwyddionprogrammeوباستخدام (EDAX) تاكدنا بان العناصر التي تم تحضيرها هي التي ظهرت بدون اي شوائب كما ان اكد تحديد المستويات البلورية عن طريق تحليل قياسات الحيود الالكتروني نتائج تحليل الاشعة السينة.

وتم در اسة الخواص المغناطيسية. وقد لوحظ ان المغنطة العظمي للعزوم لحديديات النيكل النانومتري اقل منها لحديديات النيكل في الحالة الصلبة ولوحظ ايضا ان طريقة التحضير لن تغير في الخواص المغناطيسية للعينات و ان العينات حجمها اقل من ١٠٠٠ نانومتر و  $M_{\rm s}$  and  $M_{\rm r}=0$  نانومتر و Superparamagnetic وتم حساب Bohr magneton بطريقتين (طريقة نظرية – طريقة معملية).